ROLE OF ORGANIC COMPONENTS IN SOL-GEL GLASSES

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Organic groupings play an important role for the preparation of glasses by the sol-gel route. They are responsible for the solubility of precursors, interact during the polycondensation process, the drying and ageing step and during firing. For processing, important parameters e.g. hydrolysis rates and the SiOH formation can be controlled by proper choice of the hydrolysable group. If organics remain in the glass network, remarkable changes of properties (density, hardness, refractive index) occur.

Introduction

The synthesis of non-metallic inorganic materials has become a matter of high interest during the recent three decades. Numerous authors have published results on preparation of glasses, coatings, and ceramic powders. A survey over the state of art can be obtained from the proceedings of the important congresses related to sol-gel work /1-9/. Whereas most papers are concerned with fundamental questions and laboratory scale preparations, only a few sol-gel applications are on the market, e.g. coatings /10-14/. Other industrial applications concern ceramic powders. Fundamentals are mainly investigated in the tetraethylsilicate system (hydrolysis and condensation as well as drying, ageing, and sintering) in alcoholic solutions, less with aqueous systems. Mechanisms in other systems as well as in multicomponent systems are hardly known. In all these reactions, organics are involved in various ways. If the gel derived materials are heated, organics are volatile, can decompose or are oxidized and fairly organic-free materials can be obtained. Residual organics in glassy structures can lead to a remarkable change of properties. Considering the different steps of a sol-gel preparation, the role of organics begins at a very early stage and ends with baking out during heat treatment or with the incorporation into the glass structure.

Precursors, solvents, processing

The sol-gel process in general begins in a homogeneous solu-

tion. This is desirable since it provides an almost homogeneous distribution of the single components, including low concentration components like dopants. In most cases solubility of inorganic oxides is very limited, e.g. SiO_2 , TiO_2 , ZrO_2 , and others. Organometallic precursors like alkoxides with a few exceptions provide excellent solubility in a variety of organic solvents e.g. alcohols, ketones, hydrocarbons. The organic groupings play an important role for solubility. Since homogeneity affects structure and properties of the glass, the proper choice of precursor and solvent is important. The length of the alkyl groups in alkoxides determines its reactivity with respect to hydrolysis. Figure 1 shows the rate constant for the first step (equation 1) as a function of R.

$$(RO)_3 Si(OR) + H_2 O \rightarrow (RO)_3 Si(OH) + HOR$$
 (1)

The acid catalysed hydrolysis was determined by FTIR reflection spectroscopy as a first order reaction. The data demonstrate clearly the influence of the chain length. The greatest difference appears between MeO and EtO as ligands.

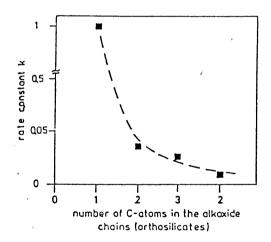


Figure 1. Relative rate constants as a function of the organic chain length in tetraalkylorthosilicates

1 = Si(OCH₃)₄; 2 = Si(OC₂H₅)₄;

3 =

solvent:ethanol; catalyst:

0.3 mmol HCl;

H₂O:Silane: 4:1.

The hydrolysis rate determines the concentration of condensable \$\int \text{SiOH}\$ groups and, together with other parameters like catalyst, the condensation rate and particle growth process. Control of \$\int \text{SiOH}\$ groups is necessary, if cocondensation of easily hydrolysable alkoxides is carried out. If water is added to a mixture of \$\text{Si(OR)}_4\$ and \$\text{Ti(OR)}_4\$, \$\text{TiO}_2\$ precipitates. Prehydrolysation of \$\text{Si(OR)}_4\$ to \$\int \text{SiOH}\$ groupings containing prepolymers and \$\text{Ti(OR)}_4\$ addition leads to homogeneous liquid systems \$/15\$/ which remain clear after water

detailed knowledge of its influence is necessary for two or more component synthesis.

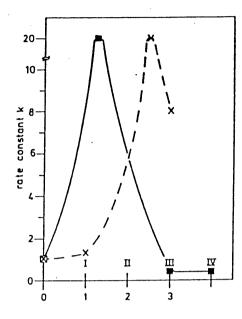


Figure 3. Hydrolysis of R' Si(OC₂H₅)_{4-n}, I, II...: n = 1; R' = CH₃ ... C₂H₅ ... X, 1, 2 ...: R' = CH₃; n = 1, 2 ... solvent: ethanol; catalyst: 0.3 mmole HCl; H₂O:silane = 4:1

Complex ligands like 8-diketones or organic acids like acetates can severely interact with alkoxides /18/. The effect on sol-gel processing is scarcely investigated. For alumina, zirconia or titania containing systems, different mechanisms take place. E.g. acetylacetonate in the case of alumina decreases condensation rate and increases the solubility of the hydrolysed species /16/. For zirconia, a drastical solubility increase can be observed /19/ and for titania, complexed with acetates, more linear polymers are obtained /20/. Complexation can be an important mean for tailoring intermediates for sol-gel processing expecially with respect to improve coatings /21/. An important finding for large piece processing was the observation that densification and shrinkage can be controlled by addition of organic materials like formamides or glycols /22/.

Organics incorporated into glass structure

In general, sol-gel processing requires a densification step at higher temperature (close to Tg) to obtain dense glasses. Organic groupings can act as network modifier, reducing network connectivity /22/ and can affect a drastical change of properties compared to inorganic glasses. If organic groupings are crosslinked, they can act as network formers. A survey over preparation techniques is given in /23/ and literature cited therein. In the

following some specific aspects of material properties due to orqanic incorporation will be discussed.

Therefore, different compositions have been investigated (I: $(C_6H_5)_n$ Si $(O_{1/2})_{4-n}$ (a), MeO₂ (b); Me = Si, Ti; II = OCH₂CH(CH₂)₃, $Sio_{3/2}$ (a); Me (III, IV) o_x (b); x = 2, 3/2, Me = Al, Ti, Zr). First of all it has to be remarked, that only in systems with very low molar contents of organics, porous gels can be obtained /22/, e.g. in I for n = 1 and a:b > 0.3:1 remarkable porosities can be determined. This is different for CH_3 as ligands where high porosities can be obtained /23/. In the system II no porosity was observed in the investigated range a:b > 1.5:1. It is remarkable since the system II contains only threedimensional crosslinking units. Since the OH content is low /24-25/ the inorganic network is nearly fixed. Taking into account that despite high concentration of inorganic units (e.g. ZrO₂ = 40 mol%, table 1, a theoretical density of ρ = 2.55 (increments: baddeleyit: 6.0; silica: 2.2; PEO: 1.2 g/cm^3) should be expected, whereas only 1.63 has been measured. It is surprising that almost no influence on n_n and only a low influence on p can be detected. An explanation can be given by the drastic influence of the organic grouping creating a high free volume compared to the organic-free inorganic compound. Whereas in the systems I and II glass temperatures have been found between room temperature and several hundred degrees /22/, in type III systems, where the epoxy-Si grouping is substituted by the methacroyl-Si /26/ and polymerized, no Tg can be detected up to the decomposition temperature of the organics (composition table 2). This leads to the conclusion that the free rotation of the type

Table 1. Densities and refractive indices of some organically modified ceramics (ORMOCERs).

molar ratio	density (g/cm ²)	n _D
90/10	1.34	_
80/20	1.37	1.517
65/35	1.45	-
90/10	1.33	-
80/20	1.40	1.508
60/40	1.63	-
90/10	1.34	_
- •		1.498
60/40	1.34	· -
	90/10 80/20 65/35 90/10 80/20 60/40 90/10 80/20	90/10

Table 2. Composition of methacroyl containing ORMOCERs.

ORMOCER system	composition molar ratio	
M Me Al M Ph Al M T M Al	45 30 25 45 30 25 75 25 75 25	

 $M = \gamma$ -methacryloxypropyltrimethoxysilane

Me = methyltrimethoxysilane

Ph = phenyltriethoxysilane

T = tetraethoxysilane

III materials is strongly hindered by additional polymerization suggesting a structure between inorganic and organic glasses. The free rotation of non-crosslinked organic groupings has been recently observed by neutron scattering techniques on condensed aminosilane systems /27/.

Mechanical properties properties of these hybrid glasses often are related to inorganic glasses. They are brittle and, compared to organic polymers, very abrasive resistant /28/. Due to the free volume causing an open structure, gas permeation is high (for H₂O ranging between two and 0.3 g/m·d at 10 μm thickness). The low values (≈ one tenth of highly water impermeable plastics) can be obtained by hydrophobizing the inorganic network as described in /29/. Structural investigations by TEM/EDAX do not show any inhomogeneities on type III compositions whereas in type I submicron structure (Figure 4) could be detected but not quantified.

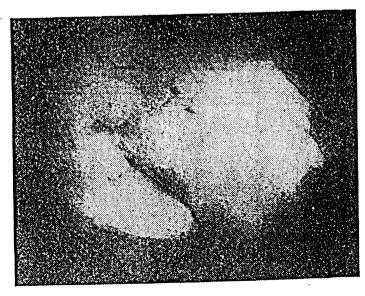


Figure 4. TEM micrograph of a modified type I gel (10 Na₂O/20 Al₂O₃/70 ($C_6^{\rm H}_5$) $_2^{\rm SiO}$)

50 nm

In this case, probably due to the different affinities of the groupings (hydrophobic $C_6^{H_5}$ versus hydrophilic \equiv SiOH and \equiv AlOH groupings) phase separation is likely on a submicron scale.

Conclusion .

The role of organics during sol-gel processing is a very important one, as a few examples have shown. Due to the concentration onto the properties of the final product, only a few data exist on this matter and the large potential has to be exploited. Organic groupings remaining in the final product change its properties drastically. These new materials (ORMOCERS) exhibit a wide variety of properties and interesting structural aspects.

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